

The Manufacture, Characterization and Aging of Novel High Temperature Carbon Fibre Composites.

Bronwyn Louise Fox

A thesis submitted for the degree of Doctor of Philosophy of
The Australian National University.



February 2001

To Dr. David Brian Fox

Statement of Originality

The PhD research program has been conducted under the principal supervision of Dr. Adrian Lowe (ANU, Department of Engineering) and Dr. Jonathan Hodgkin (CSIRO, Division of Molecular Science).

I declare that the work presented in this thesis is to my belief original, except as acknowledged in the text, and that the material has not been submitted either in whole or in part, for a degree at this or any other university.

Bronwyn Louise Fox

February 2001

ACKNOWLEDGEMENTS

First of all, I would like to thank *Adrian Lowe* and *Jonathan Hodgkin*, I couldn't have wished for two better supervisors. I would like to thank both of them for taking me on as an apprentice and for being such a pleasure to work with. Thanks to *Adrian* for all of his advice, support and for encouraging me to have a go at lecturing his Materials Science unit. Thanks also to *Marta* and *Daniel* for putting up with my frequent visits with various drafts. Thanks to *Jonathan* for his guidance over the years and for allowing me access to the wealth of experience he has in high temperature composites.

I would like to acknowledge funding from the Australian Research Council (ARC), the CSIRO Division of Molecular Science and the Australian Federal Government under the Targeted Institutional Links Scheme (TIL) administered by the Department of Employment, Education, Training and Youth Affairs (DEETYA).

I would also like to specially thank *Trevor Morton*, for his excellent advice during this project and *Buu Dao*. Both were an enormous help in setting up this study, providing materials and in training me in the black art of composite fabrication. Thanks also to *Russell Varley* for his assistance with the Dynamic Mechanical Thermal Analysis.

Thanks also to *Vincent Otieno-Alego* for being a huge help with the work using the Raman microscope and for giving me the patience to deal with highly fluorescent materials. Thanks to *Dudley Creagh* for enabling the Raman work to be completed.

Thanks to the staff and students of the Engineering Department for their help, support and friendship, particularly *Josephine Farmer*, *Pam Roberts*, *Sue Cameron*, *Shankar Kalyanasundaram*, *Mick Cardew-Hall*, *Rob Gresham*, *Bob Williamson*, *Michael Green*, *Andrew Blakers* and especially *Brenan McCarragher* for giving me the opportunity to teach "Systems Engineering Design".

I would like to *Darius Krivanek* for coming to my rescue on a number of technical matters, particularly those involving the prepregger and the autoclave.

Thanks to *James Mardel*, *David Attwood* and *Paul Compston* for helpful discussions.

I would like to acknowledge my appreciation of the very helpful staff of the Electron Microscopy Unit at the ANU, particularly *Roger Heady*, *Frank Brink* and *Sally Stowe*.

I would like to thank *Graeme Heath* from the Research School of Chemistry for assistance with the synthetic work. I would also like to thank *Horst Neumann*, *Stephen Lee* and *Nicholas Perkins* also from the RSC, for their help.

Thanks to *Ben Jar* and *Zbigniew Stachurski* for assisting in the first stage of this project.

Thank you to *Lucé Andrews* for his invaluable assistance with the writing up process.

A special thank you to *Margaret Fox* for being a fantastic role model and for encouraging me to pursue science. Thanks also to *David Fox* and *Paige Squires* for their frequent and amusing visits to Canberra.

Thank you to *Christine*, *Noel* and *Bryant Cobcroft* for their support and encouragement and for being my surrogate family in Canberra.

I owe an enormous debt to *Karen* and *James Mardel*, for their support and counselling, not to mention the culinary delights that awaited me on my frequent visits to CSIRO in Melbourne. Thanks also to *Tracey Bray* and *Heather St John* for being ever ready to join me in having Phở.

I would like to thank *Heidi Pritchard* for her boundless support, for keeping me sane (vaguely) through this whole process and for making Canberra such a fun place to live.

Finally, a very special thank you to *Tyrone Cobcroft* for accompanying me into the lab in the wee hours of the morning, providing support and for inventing spectacular ways of injuring himself just to take my mind away from it all. I couldn't have survived this far without you.

ABSTRACT

High temperature composite materials used in aerospace applications are exposed to extremely harsh conditions and must be able to withstand moisture and extremes of temperature. For example, the surface of an aircraft flying at Mach 2.4 has been estimated to reach around 177°C as a result of aerodynamic heating. This thesis has examined the effect of isothermal aging on two high temperature composite materials, a novel CSIRO composite and a commercial composite, both based on bismaleimides. Changes in mechanical properties and resin chemistry at two different temperatures were measured in order to assess the validity of accelerated aging tests.

Delamination is a major cause of failure in materials, therefore, the Mode I interlaminar fracture toughness (G_{IC}) of both materials was measured using the double cantilever beam (DCB) test. After aging at 250°C, the CSIRO CBR 320/328 composites exhibited better retention of G_{IC} than the CIBA GEIGY Matrimid® 5292 composites. After 6 weeks of aging at this temperature, the CBR 320/328 material retained 100% of its initial interlaminar fracture toughness, however the Matrimid® 5292 material retained only 64% of its initial G_{IC} . This trend was reversed at the lower aging temperature, when after 30 weeks of aging at 204°C, G_{IC} was measured at 13% of its original value for the CSIRO composites, whereas it was measured at 64% in the case of the Matrimid® composites. When the fracture surfaces of these specimens were examined using scanning electron microscopy (SEM), the commercial material was observed to show an increasing degree of porosity with aging at 204°C. It was concluded that the good property retention at the temperature, despite this observed porosity, was a result of the excellent fibre/matrix adhesion exhibited by this material.

Chemical degradation of the matrix of the composites was monitored by Fourier Transform Infrared (FTIR) and Raman Spectroscopy. Chemical changes at the core of both of these materials were found to occur concurrently with the observed changes in interlaminar fracture toughness. FTIR analysis of both matrix materials revealed the predominant degradation mechanism to be oxidation, specifically the oxidation of the methylene group bridging two aromatic rings common to the structure of both resins, was substantiated by the ingrowth of a broad peak centred at 1600 cm^{-1} . In addition to this, the pyromellitic anhydride unit present only in the CBR 320/328 composites was found to be highly resistant to the effects of aging, whereas the saturated imide, common to the cured structures of both materials, was observed to degrade.

Raman spectroscopy showed an increase in the intensity of a peak at 1646 cm^{-1} in the Matrimid® 5292 composites aged at 250°C towards the centre of the sample as a result of increased reaction of the allylic carbon-carbon double bond. At 204°C, the degree of reaction increased towards the surface of the material, possibly as a result of a reverse Diels-Alder reaction. The glass transition temperatures of both materials were found to decrease with aging, with the exception of the CSR 320/328 composites aged at 204°C, which initially increased due to continued crosslinking of the resin.

It is concluded that the degradation mechanisms at the two aging temperatures are very different. The reliability of results from accelerated (elevated temperature) aging tests has been drawn into doubt.

CONTENTS

<i>Chapter 1 – Introduction</i>	<i>1</i>
<i>Chapter 2 - Literature Review</i>	<i>5</i>
2.1 High Temperature Carbon Fibre Composites	5
2.2 Polyimides	5
2.3 Bismaleimides	8
2.4 Aging of High Temperature Composites.	11
2.4.1 Physical aging	12
2.4.2 Chemical aging of polyimide composites.	14
2.4.3 Mechanical Property Changes in Aged Polyimide Composites.	15
<i>Chapter 3 - Materials and Methods</i>	<i>17</i>
3.1 Resin Synthesis	17
3.1.1 Synthesis of CBR 320	17
3.1.2 Synthesis of CBR 328	18
3.2 Resin Degassing	18
3.3 Prepregging	19
3.4 Composite Lay-up and Cure	20
3.5 Composite Aging	22
3.6 Mode I Interlaminar Fracture Toughness Tests	22
3.7 Fourier Transform Infrared Spectroscopy	25
3.8 Raman Spectroscopy	26
3.9 Dynamic Mechanical Thermal Analysis	29
3.10 Scanning Electron Microscopy	30

<i>Chapter 4 - Isothermal Aging at 250°C</i>	32
4.1 Introduction	32
4.2 Mode I Interlaminar Fracture Toughness	32
4.2.1 Delamination Testing of Matrimid [®] 5292 Composites.....	33
4.2.2 Delamination Testing of CBR 320/328 Composites.....	33
4.2.2 Comparison of the Interlaminar Fracture Toughness Results.....	35
4.3 Scanning Electron Microscopy	37
4.4 Fourier Transform Infrared Spectroscopy	39
4.4.1 FTIR Spectroscopy of the surface of composites aged at 250°C.....	39
4.4.2 FTIR Spectroscopy of the core of Matrimid [®] 5292 composites aged at 250°C.	40
4.4.3 FTIR Spectroscopy of the core of CBR 320/328 composites aged at 250°C.	42
4.5 Raman Spectroscopy	43
4.6 Dynamic Mechanical Thermal Analysis	46
4.7 Composite Thermal Stability	47
4.8 Summary of the Results of Aging at 250°C	48
<i>Chapter 5 - Isothermal Aging at 204°C</i>	51
5.1 Introduction	51
5.2 Mode I Interlaminar Fracture Toughness	51
5.2.1 Delamination Testing of Matrimid [®] 5292 Composites.....	52
5.2.2 Delamination Testing of CBR320/328 Composites.....	52
5.2.3 Comparison of the Interlaminar Fracture Toughness Results.....	54
5.3 Scanning Electron Microscopy	55
5.4 Fourier Transform Infrared Spectroscopy	58
5.4.1 FTIR Spectroscopy of the core of Matrimid [®] 5292 composites aged at 250°C.	59

5.4.2 FTIR Spectroscopy of the core of CBR 320/328 composites aged at 250°C.	60
5.5 Raman Spectroscopy.....	60
5.5.1 Matrimid [®] 5292 composites.....	61
5.5.2 CBR320/328 composites.....	63
5.6 Dynamic Mechanical Thermal Analysis	66
5.8 Summary of the results of aging at 204°C	67
<i>Chapter 6 – Discussion</i>	<i>70</i>
6.1 Introduction	70
6.2 Composite Manufacture	70
6.3 Mode I Interlaminar Fracture Toughness.....	72
6.4 Scanning Electron Microscopy	78
6.5 Fourier Transform Infrared Spectroscopy.....	80
6.6 Raman Spectroscopy.....	84
6.7 Dynamic Mechanical Thermal Analysis	85
<i>Chapter 7 - Conclusions & further work</i>	<i>87</i>
<i>Bibliography.....</i>	<i>91</i>